



Assembly of the gas phase permeability cell

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PROJECT INFORMATION	
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Project title	Oxygen inhibition of gaseous hydrogen embrittlement
Research Program	3: Network Lifecycle Management
Milestone Report Number	2
Description	This milestone report describes the gas phase permeation cell and the remaining steps to commission the cell and first experimental studies with the cell. Hydrogen carried by gas transmission pipelines will permeate into the pipeline walls, which can lead to hydrogen embrittlement. This paper details the design of an apparatus which will be used to study the permeation of gaseous hydrogen through pipeline steels at pressures up to 17MPa and temperatures up to 200°C. The results of the permeation tests will be used to (i) characterise the diffusivity and solubility of gaseous hydrogen in pipelines, (ii) equate the conditions in hydrogen embrittlement studies which use electrochemical and gaseous hydrogen charging, and (iii) study the permeation of gaseous hydrogen blended with other gases, particularly trace amounts of oxygen. Permeation studies of hydrogen with oxygen will be used to evaluate the possible use of oxygen as an inhibitor of hydrogen embrittlement.
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Industry Proponent and Advisor Team	Tawake Rakai, General Manager – Transmission Asset Management, Tim Aujard, Head of Asset Performance, Australian Gas Infrastructure Group
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Date of approval	

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Summary of Report

Hydrogen carried by gas transmission pipelines will permeate into the pipeline walls, which can lead to hydrogen embrittlement. This paper details the design of an apparatus which will be used to study the permeation of gaseous hydrogen through pipeline steels at pressures up to 17MPa and temperatures up to 200°C. The results of the permeation tests will be used to (i) characterise the diffusivity and solubility of gaseous hydrogen in pipelines, (ii) equate the conditions in hydrogen embrittlement studies which use electrochemical and gaseous hydrogen charging, and (iii) study the permeation of gaseous hydrogen blended with other gases, particularly trace amounts of oxygen. Permeation studies of hydrogen with oxygen will be used to evaluate the possible use of oxygen as an inhibitor of hydrogen embrittlement. This report describes the gas phase permeation cell and the remaining steps to commission the cell and being experimental studies with the cell.

1. INTRODUCTION

The industry need is that gas transmission pipeline operators require knowledge about safe operating conditions for gas pipelines carrying hydrogen, in terms of the permissible safe operating pressure, and tolerable defect sizes. Otherwise, fractures are expected, with significant collateral damage.

This research will provide critical knowledge regarding the inhibition of hydrogen embrittlement (HE) by small amounts of added gases (O₂, CH₄, CO, CO₂). If 100% inhibition of HE were practical, then the mechanical behaviour of the pipeline steel for a hydrogen carrying pipeline would be exactly the same as that of a gas transmission pipeline with no hydrogen and the operating characteristics would remain the same as at the present time, regardless of the hydrogen content of the gas inside the gas transmission pipeline. The use of this strategy requires in-depth understanding because thermodynamics indicates that the inhibition is a kinetic effect.

This research assesses the feasibility of the use of gas phase inhibition of environmental hydrogen embrittlement for use in gas transmission pipelines carrying hydrogen. The research proposes (i) to study the hydrogen flux entering gas transmission pipeline steels exposed to hydrogen and hydrogen methane mixtures in the presence of small amounts of added gases such as O₂, CO, and CO₂; and (ii) study the approach to equilibrium solubility of hydrogen in the gas transmission pipeline steels, in particular how quickly equilibrium is approached.

2. BODY OF REPORT

Please see attached paper: Hoschke et al Design of a gas permeability apparatus to study oxygen inhibition of hydrogen embrittlement.

3. CONCLUSIONS

The paper describes the gas phase permeability apparatus, its design, construction, assembly and proposed use.

4. IMPLICATIONS AND RECOMMENDATIONS FOR INDUSTRY

This report is for noting.

5. NEXT STEP AND FUTURE WORK

The gas phase permeability cell as described in the attached paper is essentially complete and assembled except for some minor tubing which is to be completed Thursday 19 Oct. The hydrotest has been successfully completed, so that all required documentation has been completed for the submission at Griffith University for safety approvals for the use of the cell with hydrogen. Approval is also awaited for the use of hydrogen in the lab at Griffith Univ, the relevant documentation has been submitted. The electronics is connected, the software is in place, some fine tuning may be needed once the cell is used. Once approval is received for the use of hydrogen in the lab, calibration can be commenced for the various volumes in the apparatus to ensure accurate measurement of the egress hydrogen flux from the specimen during the permeation experiments to ensure accurate measurement of hydrogen diffusivity and hydrogen solubility.

Once calibration has been carried out, the following studies are planned. (i) Measure equilibrium solubility of hydrogen in the gas transmission pipeline steels at room temperature as a function of hydrogen partial pressure from 1 to 200 bar hydrogen in hydrogen and hydrogen methane mixtures in the presence of small amounts of up to 1% of added gases such as O₂, CO, and CO₂, for the gas pipeline steels used for the hydrogen embrittlement studies at UQ, (ii) Understand the inhibition of hydrogen entry into gas pipeline steels by the presence of the gases such as O₂, CO, and CO₂.

6. REFERENCES

(Please make sure the referencing style is consistent for all milestone reports of the project.)

Design of a gas permeation apparatus to study oxygen inhibition of hydrogen embrittlement

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Abstract—Hydrogen carried by gas transmission pipelines will permeate into the pipeline walls, which can lead to hydrogen embrittlement. This paper details the design of an apparatus which will be used to study the permeation of gaseous hydrogen through pipeline steels at pressures up to 17 MPa and temperatures up to 200 °C. The results of the permeation tests will be used to (i) characterise the diffusivity and solubility of gaseous hydrogen in pipelines, (ii) equate the conditions in hydrogen embrittlement studies which use electrochemical and gaseous hydrogen charging, and (iii) study the permeation of gaseous hydrogen blended with other gases, particularly trace amounts of oxygen. Permeation studies of hydrogen with oxygen will be used to evaluate the possible use of oxygen as an inhibitor of hydrogen embrittlement.

Index Terms—hydrogen embrittlement, gas permeation, pipeline steel, oxygen inhibition

I. INTRODUCTION

There has been a large amount of research into hydrogen embrittlement (HE) of pipeline steels. HE in pipelines could be caused by electrochemically generated hydrogen, from sources such as corrosion and cathodic protection [1]–[4], or by gaseous hydrogen within the pipeline. Much of the existing research is motivated by the prospective use of existing steel pipelines to transport gaseous hydrogen, which, when produced from renewable energy sources, would aid in decarbonisation [5]–[7].

Electrochemical charging is frequently used to introduce hydrogen into test specimens when studying HE [8]–[12]. Electrochemical testing can be related to HE caused by gaseous hydrogen provided the relationship between the electrochemical charging conditions and gaseous hydrogen conditions is known. This relationship has been investigated by characterising the hydrogen fugacity in both environments, which can be found using hydrogen permeation experiments or a method of measuring the hydrogen content such as Thermal Desorption Spectroscopy [13]–[16]. In particular, Koren et al. [17] quantified the relationship between gaseous and electrochemical hydrogen charging for X65 pipeline steel, using 3.5% NaCl for electrochemical charging. Data are lacking for other grades of pipeline steels and other electrolytes.

There is compelling evidence that gaseous HE in metals can be significantly inhibited in hydrogen containing trace amounts of certain gases, particularly oxygen [18]–[21]. This effect has been observed in steels similar to pipeline steels [22]–[25]. This phenomenon is attributed to the preferential adsorption of the inhibiting gas preventing the adsorption of hydrogen to the metal surface, as shown using surface imaging techniques by Germer and MacRae [26] and Holbrook et al. [22], [27].

However, the influence of oxygen on hydrogen diffusion has not been investigated. Ewing and Ubbelohde [28] investigated hydrogen permeation in the presence of oxygen, but their work focused on oxygen inhibiting the desorption of hydrogen from the exit side of a permeation specimen, and as such was not able to study hydrogen diffusion in the metal nor the subsurface hydrogen concentration.

Oxygen inhibition of HE is an attractive strategy for natural gas high-pressure pipelines transporting hydrogen. Is this a feasible strategy on a timescale of tens of years? A deeper quantitative (i.e. timescale, effectiveness, etc.) and qualitative (i.e. type of gases) understanding of gas phase inhibition of HE is needed. The following will provide a deeper understanding of gas phase inhibition of HE.

- 1) A gas permeation apparatus (GPA) is needed that allows study of the hydrogen flux through a pipeline steel specimen subject to high pressure hydrogen containing a known quantity of added gases, such as O₂.
- 2) A model of the GPA is needed to extract the hydrogen solubility and diffusivity from the measurement of the pressure build-up in the downstream volume.
- 3) Gas phase permeation experiments need to be carried out with the GPA using pure hydrogen to study the hydrogen solubility and diffusivity of the steel as a function of hydrogen pressure and temperature. This allows gaseous hydrogen charging to be equated to electrochemical charging, and provides a base for comparison with subsequent experiments with added gases. Similar experiments are needed with different gas mixtures to study the influence of the added gas, including the dependence on partial pressure, temperature, steel grade, type and amount of added gas.
- 4) Measurement of the hydrogen concentration using Ther-

mal Desorption Spectroscopy (TDS) is needed to confirm the GPA data.

- 5) Use of surface characterisation techniques such as XPS is needed to study the surface mechanisms involved in these inhibition effects.

This paper addresses the first two objectives. The gas-phase permeation apparatus design is described in the following section. An overview over the samples (materials and gases) is given thereafter.

II. APPARATUS DESIGN

Permeation experiments study the permeation of hydrogen through a membrane of the material of interest. The transport parameters (diffusivity, solubility, and permeability) are evaluated from the hydrogen flux through the specimen.

Typical hydrogen permeation methods include methods that use (i) electrochemical hydrogen charging and detection, (ii) gas phase charging and detection, and (iii) gas phase charging with electrochemical detection [29]. In a typical permeation experiment, the inlet hydrogen flux (as determined by the current or pressure) is quickly increased or decreased, and the exit hydrogen flux is recorded over time. The inlet flux over time is represented by a step function, while the exit flux resembles a sigmoid function. A single change in the inlet flux, and the corresponding change in the exit flux, is referred to as a permeation transient [30]. Multiple transients can be performed successively to collect multiple data points for the steel's properties.

Further distinction between hydrogen permeation methods can be made by the governing boundary conditions [31]. In an ideal case, the charging condition (e.g. pressure) remains constant and a steady-state flux is established. In contrast, in a gas phase permeation rig, a fixed volume on the downstream side of the permeation membrane results in downstream pressure increasing over time, and a fixed upstream volume results in a decreasing upstream pressure. This causes the flux to be in a transient state throughout the experiment.

This proposed gas permeation apparatus (GPA) uses high-pressure hydrogen gas charging, closely replicating the pressures expected in gas pipelines carrying hydrogen. The approach enables the study of the influence of different hydrogen gas blends and added impurities by injection of the proposed gas phase inhibitors. Finally, gas phase permeation allows the determination of the hydrogen solubility in the steel membrane as a function of the hydrogen partial pressure.

A schematic of the GPA is presented in Fig. 1. Table I gives an overview of the sections and their associated valves. There are gas inlets for hydrogen, methane, oxygen, and carbon monoxide. To create a gas mixture, each constituent gas is fed into one of the two sample cylinders ($V_{ref,1}$ and $V_{ref,2}$) at a desired pressure. After the first cylinder is charged, the manifold is evacuated such that the second cylinder is not contaminated. Once both cylinders are charged, Valve 7a and Valve 7b are opened and the gas mixture is created in the manifold and the two sample cylinders. This produces a volume $>300\text{ cm}^3$ from which the upstream section can be charged multiple times.

TABLE I: Sections and partitioning of the GPR. Numbers refer to valve numbers in Fig. 1 (not including pressure relief valves as they don't serve control purpose).

GROUP	NO.	SECTION	ENCLOSED BY
Supply	1	Metal-Hydride Sample Cylinder	1d
	2	Hydrogen Inlet	1a, 1b, 1d
	3	Oxygen Inlet	2a, 2b
	4	Carbon Monoxide Inlet	3a, 3b
	5	Methane Inlet	4a, 4b
Manifold	6	Manifold	1b, 2b, 3b, 4b, 5, 6, 7a, 7b, 9
	7a	Reference Volume 1	7a
	7b	Reference Volume 2	7b
Upstream	8a	Upstream High Pressure	9, 11a, 11b, 12a
	8b	Upstream Permeation Cell	11a, Cell
	8c	Upstream Low Pressure	11b
Downstream	9a	Downstream Manifold	Cell, 13a, 13b, 13c
	9b	Downstream Low Pressure	13b
	9c	Downstream High Pressure	13c, 14
Vacuum	10	Vacuum Outlet	6, 12b, 14

The upstream pressure can be precisely controlled to a desired pressure using a needle valve (Valve 9).

The permeation cell holds the disc specimen as shown in Fig. 3. At the beginning of each experiment, a vacuum is created downstream of the specimen, and high-pressure gas is introduced into from the upstream section. This can result in pressure gradients across the specimen of up to 200 bar. The minimum thickness of a circular plate with fixed edges is calculated using [32]

$$t = 2r \sqrt{\frac{0.19P}{\sigma}} \quad (1)$$

where r is the radius of the disc, P is the applied pressure and σ the allowable stress. A steel disc specimen for testing in the GPA has a thickness of 4 mm and an exposed radius of 11 mm, which results in the maximum stress remaining below half of the yield strength in the softest steel to be tested (X42, 290 MPa). Typically, hydrogen does not significantly reduce the yield strength of steel [6], [33]. For a pure annealed iron specimen, the upstream pressure may not exceed 50 bar.

The hydrogen flux is evaluated from the pressure increase in the downstream vacuum section using a high-accuracy pressure transmitter (MKS Baratron AA04A). Additionally, the specimen temperature can be controlled by a band heater attached to the permeation cell, allowing the study of the influence of temperature on hydrogen diffusion and solubility. A vacuum pump is used to evacuate the system, and a helium inlet line is used to purge the system. The design of the GPA allows for evacuating specific sections (e.g. manifold for gas mixing) or the whole system (i.e. evacuation before or after individual runs).

The maximum pressure and temperature ratings are dictated by the weakest link. All tubing, piping, and the double-ended sample cylinders are rated ≥ 344 bar at room temperature. The maximum temperature is 537°C . Diaphragm valves are rated 209 bar at 65°C , whereas high purity diaphragm valves, used in close proximity to the permeation cell, may reach pressures of up to 241 bar at room temperature and 192 bar at 121°C . Most sections only experience room temperature. To stay within

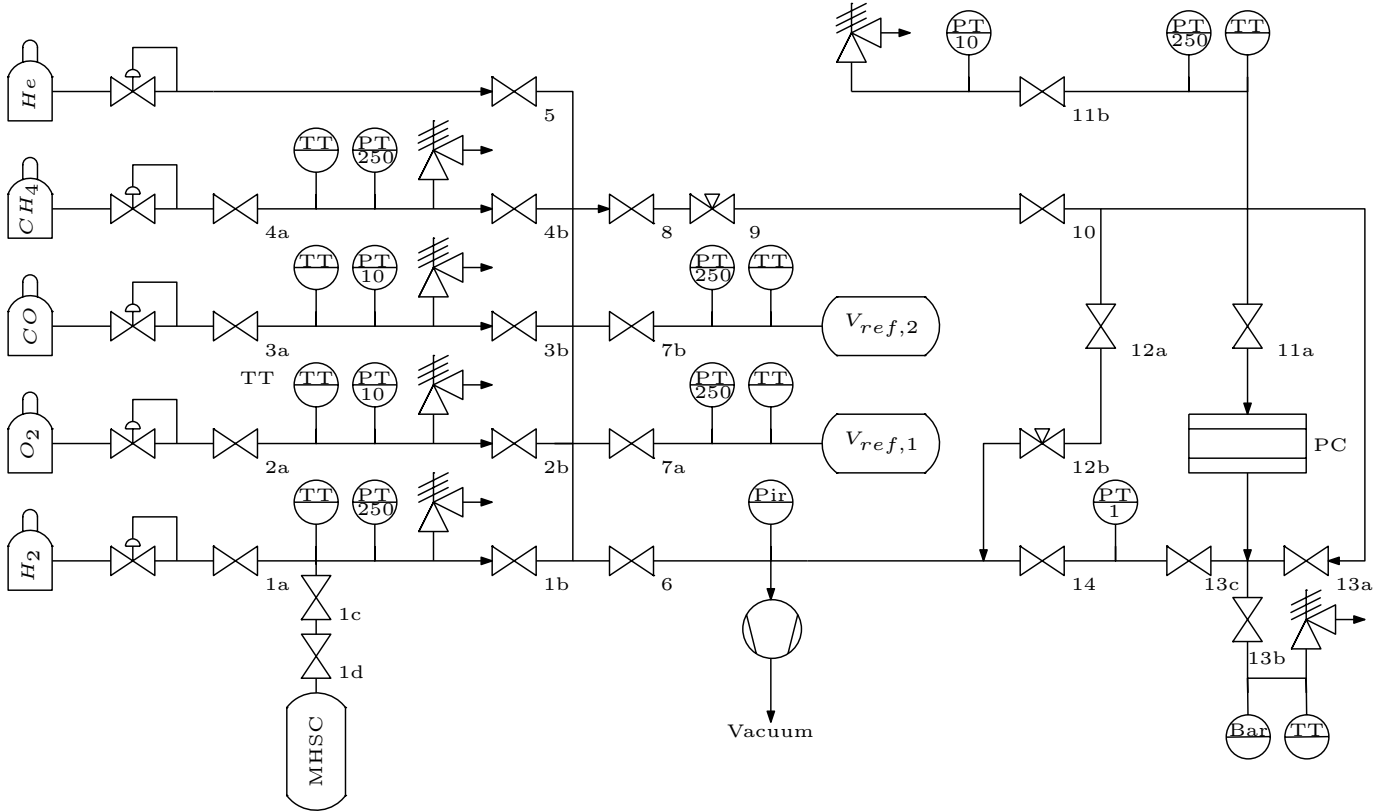


Fig. 1: Piping and Instrumentation Diagram of the GPR. The integrated thermocouples in the permeation cell are not drawn. Arrows depict the main directions of gas flow. Where different directions are possible, no arrows are shown.

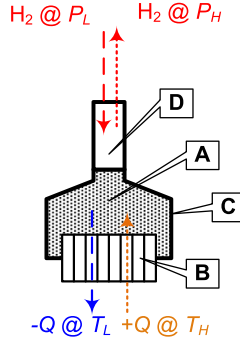
these limitations, only final pressures of approximately 120 bar to 150 bar, depending on the gas blend, are feasible. After mixture preparation in sections 7, 8 and 9, gas is expanded into the upstream compartment. While pressures in the mentioned mixing sections may only safely reach 209 bar, the pressure after expansion decreases, which conforms to the temperature-pressure rating of the surrounding valves.

At the beginning of each experiment, gases are sourced from cylinders through entry sections (sections 3, 4 or 5 in Table I). Hydrogen, as the main constituent of every gas mixture, is an exception, as hydrogen is supplied by a metal hydride sample cylinder (MHSC). In order to assess the effects of added gases for each experiment, the sourced hydrogen needs to contain as little impurities as possible. Usually, hydrogen fed from conventional gas cylinders for laboratory use does not meet purity requirements. Furthermore, gas cylinder pressures may be too low. An MHSC effectively functions as a compressor by absorbing hydrogen at low temperatures and desorbing it upon heating (see fig. 2). In a laboratory, an MHSC acts as a good alternative to a conventional compressor or gas cylinder. A single-stage MHSC can be used for periodic charging and discharging, and does not involve any moving parts. Additionally, hydrogen gas can be reused between experiments by cooling and reheating the MHSC, and, due to the closed design, contamination is less likely inside the apparatus. Furthermore, desorption of hydrogen from the metal

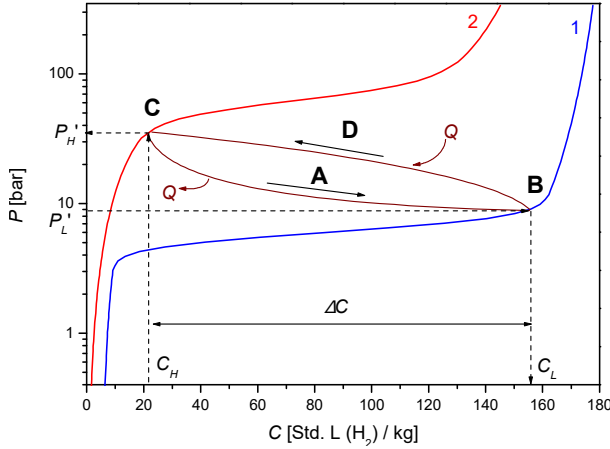
hydride produces high-purity gas, suitable for permeation studies. For a given temperature, absorption is driven by the difference between equilibrium pressure and supply pressure. Likewise, desorption is driven by the difference between equilibrium pressure and discharge pressure. The equilibrium pressure is described by the van 't Hoff equation [34]–[36]

$$\ln P_{eq} = \frac{\Delta H}{RT} - \frac{\Delta S}{R} \quad (2)$$

where P_{eq} is the equilibrium pressure at temperature T , ΔH and ΔS are material-dependent enthalpy and entropy of hydride formation with respect to a standard state, and R is the universal gas constant. This equation is an ideal approximation, ignoring the hysteresis that occurs between absorption and desorption. Fig. 2b indicates that hydrogen is produced at an almost constant pressure. Isotherm 1 is a lower temperature (T_L) and isotherm 2 is a higher temperature (T_H). When heat is rejected from the MHSC, the equilibrium pressure decreases. As long as the supply pressure exceeds the pressure at the lower operating point B (P_L , T_L), the MHSC is charged, forming metal hydride. Vice versa, the decomposition takes place when heating to a higher operating point C (P_H , T_H). The available amount of hydrogen equals the change of concentration Δc between points B and C and is called the reversible storage capacity [36]. Identifying the operating pressures, temperatures, and storage capacity allows for the selection of a suitable material. Here LaNi_5 is used. Heat is applied using a heat gun. The



(a) Layout of a single-staged periodically operated compressor, with metal hydride (A), heat source/sink (B), pressure vessel (C) and piping including a gas filter (D). The dashed arrow corresponds to path A in Fig. 2b, whereas the dotted arrow equates to path D.



(b) Schematic illustration of two pressure-composition isotherms for a metal hydride. Isotherm 1 is at 20 °C, and isotherm 2 is at 150 °C. Paths A and D represent cyclical absorption and desorption where heat is removed or supplied, respectively. Points B and C depict equilibria states.

Fig. 2: The design and operation of metal hydride compressors. Reproduced from Lototsky et al. [35] with permission.

temperature-pressure rating of the MHSC vessel is 210 bar at 260 °C, which is higher than the high operating point.

III. ENVIRONMENTS AND MATERIALS

Annealed pure iron is to be used for initial test runs on the GPA. The transport parameters of iron are readily available in the literature, which makes it easier to assess the accuracy of the set-up. In subsequent experiments three different pipeline steels (X42, X65 and X70) will be used as permeation membranes to study their hydrogen solubility and diffusivity when charged by high-pressure hydrogen. Sample materials will be cut into discs with a diameter of 32 mm and a thickness of 4 mm to be loaded into the permeation cell.

The gas mixtures are summarised in Table II. Permeation tests in pure hydrogen are expected to produce the highest diffusion coefficients. Therefore, they represent a baseline for gas mixtures that are assumed to inhibit hydrogen diffusion. Gas transmission pipelines that are to be converted to carry

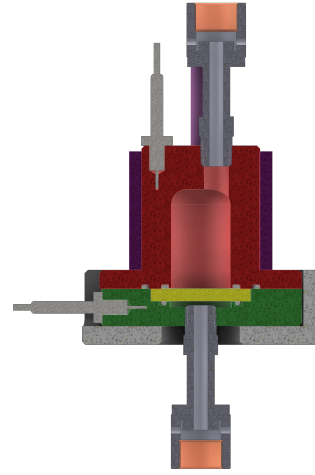


Fig. 3: Detailed drawing of the permeation cell. The disc specimen is coloured yellow. The housing (red and green) is made of stainless steel and enclosed by a band heater (purple) to control the specimen's temperature. Ceramic insulation (dark grey) limits heat loss and steep temperature gradients across the permeation cell.

pure hydrogen will exhibit these conditions. Methane-hydrogen mixtures of various hydrogen contents are investigated as they represent prospective operating conditions where hydrogen is blended into natural gas pipelines. In later experiments small amounts of oxygen or carbon monoxide will be blended into pure hydrogen. These experiments will study the feasibility of the use of these gases as hydrogen embrittlement inhibitors. Future investigation of mixtures including other gases is possible by simply switching the gas inlet connections to different gas cylinders.

Hydrogen has a lower explosive limit (LEL) of 4 vol.% and an upper explosive limit (UEL) of 75 vol.% while methane's LEL is 5 vol.% and UEL is 15 vol.%. All sample mixtures are outside these ranges and are thereby too lean to be flammable.

IV. DATA ANALYSIS

Consider that the specimen is exposed to pure hydrogen in the upstream chamber, and that the upstream hydrogen pressure is increased from 0 MPa to 10 MPa essentially instantaneously

TABLE II: Investigated gas mixtures. Target compositions in volume fraction and partial pressure are listed. Note that 0.001 vol.% is equal to 10 vppm.

MIXTURES	H ₂		CH ₄		O ₂		CO		P_{Total} (bar)
	vol.%	bar	vol.%	bar	vol.%	bar	vol.%	bar	
Pure H ₂	100	170	0	0	0	0	0	0	170
CH ₄ based	5	7.5	95	142.5	0	0	0	0	150
	25	37.5	75	112.5	0	0	0	0	150
O ₂ based	99.99	149.985	0	0	0.01	0.015	0	0	150
	99.999	119.9988	0	0	0.001	0.0012	0	0	120
CO based	99.99	149.985	0	0	0	0	0.01	0.015	150
	99.999	119.9988	0	0	0	0	0.001	0.0012	120

by opening the upstream valve. The hydrogen flux, J , has units $\text{mol/m}^2/\text{s}$ and is given by

$$J(t) = \frac{dn}{dt} \frac{1}{A} \quad (3)$$

where n is the amount of hydrogen, t is time, and A is the area from which hydrogen is egressing. If the boundary conditions are that of constant hydrogen concentration on the entry side, and 0 hydrogen concentration on the exit side, the flux is expected to be given by [30], [37]

$$\frac{J_t - J_0}{J_\infty - J_0} = \frac{2L}{\sqrt{\pi Dt}} \sum_{n=0}^{\infty} \exp\left(-\frac{(2n+1)^2 L^2}{4Dt}\right) \quad (4)$$

where J_0 is the flux before the transient (in this case 0), J_∞ is the steady-state value which the flux approaches, L is the thickness of the specimen, D is the diffusion coefficient, and t is the time since the transient began. The pressure, P , in the downstream chamber is related to the flux by

$$\frac{dP}{dt} = J(t) \frac{AZRT}{V} \quad (5)$$

where Z is the compressibility factor, R is the gas constant, T is the temperature, and V is the downstream volume. Combining (4) and (5) gives

$$\frac{\left(\frac{dP}{dt}\right)_t - \left(\frac{dP}{dt}\right)_0}{\left(\frac{dP}{dt}\right)_\infty - \left(\frac{dP}{dt}\right)_0} = \frac{AZRT}{V} \frac{2L}{\sqrt{\pi Dt}} \sum_{n=0}^{\infty} \exp\left(-\frac{(2n+1)^2 L^2}{4Dt}\right) \quad (6)$$

Fitting the shape of this theoretical pressure transient to the experimentally measured time derivative of the pressure enables measurement of the effective diffusion coefficient of the steel. The hydrogen concentration in the specimen in equilibrium with the applied hydrogen pressure is derived from Fick's first law of diffusion and is given by [38]

$$C_H = \frac{J_\infty L}{D} \quad (7)$$

$$C_H = \frac{\left(\frac{dP}{dt}\right)_\infty V \cdot L}{AZRT \cdot D} \quad (8)$$

Equations 6 and 8 can be used to determine the effective diffusivity and hydrogen concentration in the steel. Similarly, consider that the upstream chamber contains hydrogen with 1000 vppm oxygen. Consider increasing the pressure essentially instantaneously from 0 bar to 100 bar. The oxygen is expected to inhibit the ingress of hydrogen into the specimen. The resulting hydrogen permeation transient can be fitted to Equation (6) and used to predict the hydrogen diffusion properties when inhibited by oxygen, including the time to steady-state flux, $t_{\infty, O}$ and the hydrogen concentration, $C_{H, O}$. The quantities $t_{\infty, O}/t_\infty$ and $C_{H, O}/C_H$ can be used to quantify the effectiveness of the oxygen inhibition of gaseous hydrogen embrittlement.

As permeation in the GPA occurs, a small amount of hydrogen is transferred from the upstream section to the downstream section, resulting in the upstream pressure slightly decreasing and the downstream pressure slightly increasing. This causes the entry and exit concentrations to vary over time.

As such the boundary conditions differ from those given in McBreen, Nonis and Beck [30], and Equations (6) and (8) are not entirely accurate to gaseous permeation. However, the error is expected to be small. In future work we plan to evaluate this error, and derive new governing equations from these varying boundary conditions.

V. SUMMARY

The gas permeation apparatus will be used to study hydrogen permeation through pipeline steels. Experiments with the high-pressure hydrogen gas can be equated to permeation studies using cathodic charging by correlating the subsurface hydrogen concentrations [13], [21], [39]. Gas impurities, including oxygen, will be added to the hydrogen to study their influence on hydrogen diffusion, allowing assessment of the feasibility of gas-phase inhibition of HE to protect steel pipelines carrying hydrogen.

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